SYNTHESIS AND PROPERTIES OF ORDERED PEROVSKITES  ${\rm A_2\,(Ln^{3+}M^{5+})\,O_6}~({\rm A=Ba\,,Sr,\,Ln=Y\,,Gd\,,Dy\,,Er,\,M=Mo\,,W)^{\,1}})$ 

Kiichiro KAMATA, Masahiro YOSHIMURA, Tetsuro NAKAMURA and Toshiyuki SATA

Research Laboratory of Engineering Materials,

Tokyo Institute of Technology, Ookayama, Tokyo

Ordered perovskites,  $Ba_2(LnMo)O_6$ ,  $Sr_2(LnMo)O_6$  and  $Ba_2(LnW)O_6$ , were newly synthesized. The crystallographic, electrical and magnetic properties for these compounds were investigated. The magnetic moments of  $Mo^{5+}$  and  $W^{5+}$  were determined. Magnetic properties indicated that the interaction between  $Ln^{3+}(4f^n)$  and  $Mo^{5+}(4d^1)$  or  $W^{5+}(5d^1)$  was fairly weak in these compounds.

Several ordered perovskites  $A_2(B'B'')O_6$  which contain two paramagnetic cations in the B site of the perovskite lattice have been synthesized  $^{2-5}$ ). These compounds have interesting properties, a ferrimagnetism and a metallic conduction in  $Sr_2(FeMo)O_6^{2}$  and  $Ba_2(FeRe)O_6^{3}$ , a ferrimagnetism and a semiconduction in  $Ba_2(MnRe)O_6^{3}$ ,  $Sr_2(CrMo)O_6$  and  $Sr_2(CrW)O_6^{4}$ . In these reports the interaction between  $3d^n$  electrons and  $4d^1(Mo^{5+})$  or  $5d^1(W^{5+},Re^{6+})$  electrons were studied.

We tried to prepare new perovskites,  $A_2$  (LnMo)O $_6$  and  $A_2$  (LnW)O $_6$  with A=Ba,Sr and Ca, in order to study the magnetic interaction between  $4f^n$  (Ln $^{3+}$ ) and  $4d^1$  (Mo $^{5+}$ ) or  $5d^1$  (W $^{5+}$ ). Lanthanide ions,  $Gd^{3+}$  ( $4f^7$ ),  $Dy^{3+}$  ( $4f^9$ ) and  $Er^{3+}$  ( $4f^{11}$ ) with large magnetic moments and  $Y^{3+}$  with no magnetic moment, were chosen for B' cations in this work.

Appropriate mixtures of reagent-grade  ${\rm BaCO_3}$  (or  ${\rm Ba\,(NO_3)_2}$ ),  ${\rm SrCO_3}$ ,  ${\rm CaCO_3}$ ,  ${\rm Y_2O_3}$ ,  ${\rm Gd_2O_3}$ ,  ${\rm Dy_2O_3}$ ,  ${\rm Er_2O_3}$ ,  ${\rm MoO_3}$  and  ${\rm WO_3}$  were heated at 1200-1300°C for 2hr, and reground and reheated at 1250-1350°C for 5hr in the atmosphere of wet hydrogen. The oxygen partial pressure in this atmosphere was determined to be  $10^{-14}{\rm atm}^6$ ). X-ray powder diffraction analysis showed that only the compounds  ${\rm Ba_2\,(LnMo)O_6}$ ,  ${\rm Sr_2\,(LnMo)O_6}$ , and  ${\rm Ba_2\,(LnW)O_6}$  had been synthesized with perovskite structures.  ${\rm Ba_2\,(LnMo)O_6}$  and  ${\rm Sr_2\,(LnMo)O_6}$  were black and  ${\rm Ba_2\,(LnW)O_6}$  was deep blue.

X-ray powder diffraction patterns were taken at room temperature using  $CuK\alpha$  radiation. Super-structure lines showing the alternating arrangement in B site were observed in  $Sr_2(LnMo)O_6$  and  $Ba_2(LnW)O_6$ , but not observed in  $Ba_2(LnMo)O_6$ . The d.c. conductivities were measured over a temperature range from 80 to 600K. The measurements were made by the four-probe method for ceramic samples in a stream of hydrogen. The magnetic susceptibilities were measured using a magnetobalance in a temperature range from 80 to 300K.

The crystallographic, electrical and magnetic properties of the synthesized perovskites are summarized in Table I.

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Table	I.	Physical	properties	for	$A_2$ (LnM)	,,)o <sup>6</sup>	compounds	at	room	temperature

Compound	Tolerance factor	Symmetry	Lattice constant (A)	Conductivi- ty (Ωcm)-l	Activation energy *	$P_{\mathrm{eff}}^{\star\star} (\mu_{\mathrm{B}})$	P_Ln (µ <sub>B</sub> )	asympt. Curie temp. (K)
Ba <sub>2</sub> YMo0 <sub>6</sub>	0.98	cubic	8.388	1.6×10 <sup>-2</sup>	0.15	1.5	0	-110
Ba2GdMo06	0.97	cubic	8.452	2.8×10 <sup>-2</sup>	0.13	7.8	8.0	+5
Ba <sub>2</sub> DyMo0 <sub>6</sub>	0.98	cubic	8.406	1.1×10 <sup>-2</sup>	0.13	10.4	10.6	-13
Ba <sub>2</sub> ErMoO <sub>6</sub>	0.98	cubic	8.376	7.3×10 <sup>-2</sup>	0.11	9.5	9.5	-10
Sr <sub>2</sub> YMoO <sub>6</sub>	0.93	p.cubic	8.180	1.7×10 <sup>-2</sup>	0.13	1.5	0	-100
Sr <sub>2</sub> GdMoO <sub>6</sub>	0.92	p.cubic	8.260	2.3×10 <sup>-2</sup>	0.12	7.8	8.0	-3
Sr <sub>2</sub> DyMoO <sub>6</sub>	0.93	p.cubic	8.230	9.3×10 <sup>-3</sup>	0.13	10.4	10.6	-10
Sr <sub>2</sub> ErMoO <sub>6</sub>	0.93	p.cubic	8.132	3.8×10 <sup>-2</sup>	0.12	9.5	9.5	-12
Ba <sub>2</sub> YWO <sub>6</sub>	0.98	cubic	8.378	1.5×10 <sup>-7</sup>	0.2-0.3	0.96	0	<b>-</b> 70
Ba <sub>2</sub> GdW0 <sub>6</sub>	0.97	cubic	8.431	1.5×10 <sup>-7</sup>	0.2-0.3	7.5	8.0	+4
Ba <sub>2</sub> DyWO <sub>6</sub>	0.98	cubic	8.392	2.5×10 <sup>-7</sup>	0.2-0.3	10.4	10.6	-13
Ba <sub>2</sub> ErWO <sub>6</sub>	0.98	cubic	8.366	5.0×10 <sup>-7</sup>	0.2-0.3	9.4	9.5	-10

<sup>\*</sup> from 200 to 500K. 7)

\*\* Effective magnetic moment observed: Diamagnetism were corrected according to Selwood?

\*\*\* Effective magnetic moment for lanthanide ion: from Kittel,

"Introduction to Solid State Physics", John Wiley & Sons, (1967) 3rd Ed., p.437.

The lattice constants observed in these perovskites were closely related to the ionic radii of constituent ions. The lattice constants for  $Ba_2(LnMo)O_6$  were in agreement with the previous values<sup>5)</sup>. The difference between the lattice constant of  $Ba_2(LnMo)O_6$  and that of  $Ba_2(LnW)O_6$  should result from the difference between the ionic radius of  $Mo^{5+}$  and that of  $W^{5+}$ which was absent in recent works<sup>9-10)</sup>. The plot of cell volumes vs. ionic radii(Fig. I) gave  $O.625\mathring{A}$  for the radius of  $W^{5+}$ . The

tolerance factors  $^{8)}$  for the perovskites  $\mathrm{Ba}_2(\mathrm{LnW})_{_{_{\!\!6}}}$  were calculated using this value. The tolerance factors were 0.97-0.98 for cubic perovskites and 0.92-0.93 for pseudocubic perovskites. The value(0.89) estimated for Ca compounds suggests a difficulty in the formation of the perovskite phase.

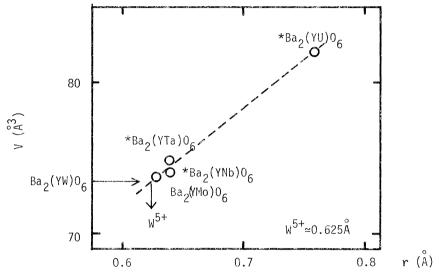


Fig. I. Plot of the cell volumes  $\mathrm{Ba}_2(\mathrm{YM}^{5+})0_6$  vs. the ionic radii of  $\mathrm{M}^{5+}$  ions after Shannon  $^{9-10}$ . \* the cell volume derived from Galasso  $^{11}$ ).

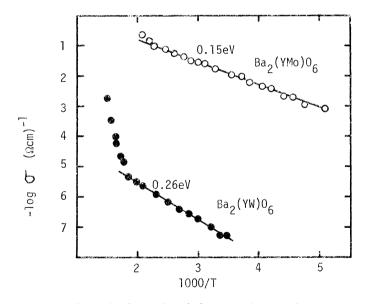


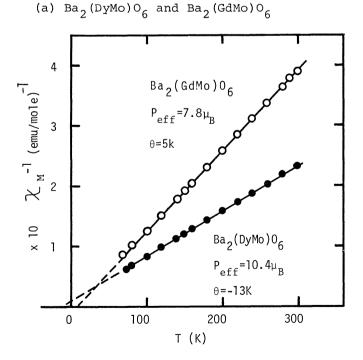
Fig. II. Electrical conductivity vs. temperature

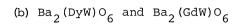
Fig. II shows the electrical conductivities of  $Ba_2(YMO)O_6$  and  $Ba_2(YW)O_6$ . Similar semiconducting behaviors were observed in the other perovskites obtained. The electrical conductivities at room temperature,  $10^{-2}$ - $10^{-3}(\Omega cm)^{-1}$ , and their activa-

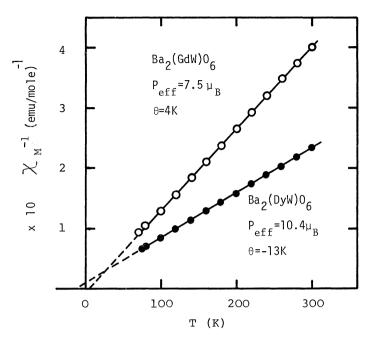
tion energies, 0.1-0.2eV(200-500K), were observed in the perovskites containing Mo<sup>5+</sup>,  $A_2$  (LnMo)O<sub>6</sub>. Similarly, the values,  $10^{-6}$ - $10^{-7}$  ( $\Omega$ cm)<sup>-1</sup> and 0.2-0.3eV, respectively, were observed in those containing W<sup>5+</sup>,  $A_2$  (LnW)O<sub>6</sub>. From these results it can be said that perovskites  $A_2$  (LnMo)O<sub>6</sub> are better electrical conductors than perovskites  $A_2$  (LnW)O<sub>6</sub>.

All perovskites obtained showed paramagnetic behaviors obeying the Curie-Weiss law. Fig. III shows the reciprocal susceptibility vs. temperature relations of several perovskites. No Neel points of them were observed over a temperature range from 80 to 300K. The magnetic moment  $0.96\mu_B$  of  $Ba_2(YW)O_6$  is in good agreement with that of  $SrLa(MgW)O_6^{-6}$ ; therefore the magnetic moment should be attributed to  $W^{5+}$  ion. The magnetic moment  $1.5\mu_B$  of  $Ba_2(YMO)O_6$  and  $Sr_2(YMO)O_6$  might be attributed to  $MO^{5+}$  ion. Since the effective moment was calculated by the formula:  $P_{eff} = \sqrt{P_{Ln}^2 + P_{M5+}^2}$ , the contribution of  $P_{M5+}$  was relatively small on the total moment  $P_{eff}$ . When  $P_{Ln}$  was large in particular, the magnetic moment of each perovskite became nearly equal to that of the lanthanide ion as seen in Table I. The values of the asymptotic Curie temperature for all perovskites were very small. From above results it can be deduced that the magnetic interaction between  $Ln^{3+}(4f^n)$  and  $MO^{5+}(4d^1)$  and  $W^{5+}(5d^1)$  is very weak in these compounds.

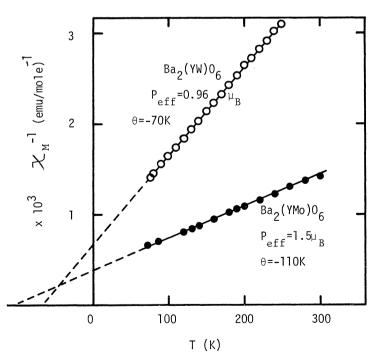
Fig. III Reciprocal susceptibility vs. temperature







## (c) $Ba_2(YMO)O_6$ and $Ba_2(YW)O_6$



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